

# Patient-Specific Finite-Element Analyses of the Proximal Femur with Orthotropic Material Properties Validated by Experiments

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*Patient-specific high order finite-element (FE) models of human femurs based on quantitative computer tomography (QCT) with inhomogeneous orthotropic and isotropic material properties are addressed. The point-wise orthotropic properties are determined by a micromechanics (MM) based approach in conjunction with experimental observations at the osteon level, and two methods for determining the material trajectories are proposed (along organs outer surface, or along principal strains). QCT scans on four fresh-frozen human femurs were performed and high-order FE models were generated with either inhomogeneous MM-based orthotropic or empirically determined isotropic properties. In vitro experiments were conducted on the femurs by applying a simple stance position load on their head, recording strains on femurs' surface and head's displacements. After verifying the FE linear elastic analyses that mimic the experimental setting for numerical accuracy, we compared the FE results to the experimental observations to identify the influence of material properties on models' predictions. The strains and displacements computed by FE models having MM-based inhomogeneous orthotropic properties match the FE-results having empirically based isotropic properties well, and both are in close agreement with the experimental results. When only the strains in the femoral neck are being compared a more pronounced difference is noticed between the isotropic and orthotropic FE result. These results lay the foundation for applying more realistic inhomogeneous orthotropic material properties in FEA of femurs. [DOI: 10.1115/1.4004180]*

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## 1 Introduction

Patient-specific finite element (FE) analyses of the human femur are widely used nowadays to predict its mechanical response (see, e.g., Refs. [1–4], and references therein). The validity of the FE results depends largely on an accurate description of bone's geometry and proper assignment of material properties to the FE model [5–8]. The generation of the bone's geometry (and thereafter the FE mesh) is considered, in large extent, as a solved problem [6,8–11]. The proper assignment of material properties on the other hand, is still under active research because of the inherent inhomogeneous and anisotropic nature of bone's tissue (in addition to the homogenization process). Most past FE studies assumed the bone to be inhomogeneous isotropic [1,3,5,7,10,12–14] due to simplicity and the limited knowledge of the anisotropic behavior. Based on the relationship between CT numbers, Hounsfield Units (HU) and a density measure  $\rho$ , the inhomogeneous Young's modulus  $E[\rho(HU)]$  has been estimated empirically [15,16]. Many empirical  $E(\rho)$  relations are reported and their influence on FE-predictions was widely investigated (see, e.g., Ref. [3]). The isotropic strategy has also been motivated by limited comparative studies claiming that the assigned orthotropic material model has a minor influence on the FE result compared to the isotropic one. Peng et al., for example, compared FE models in using orthotropic and inhomogeneous isotropic properties (although no experimental data were

presented) and concluded that the differences are negligible [13]. Baca et al. compared FE models of the proximal femur and “small” specimens with either isotropic or orthotropic inhomogeneous properties [17]. The reported results show a significant difference for small bone specimens, whereas in the entire organ (proximal femur), the difference is negligible. A recent numerical study [18] concluded that the differences between the two property assignments are more significant (maximum differences in Von Mises stresses about 13%) in some local region. This conclusion is limited since in Ref. [18] only one bone specimen was modeled and no comparison to experiments was conducted.

Although some empirical relations between the orthotropic constants and bone density have been suggested [19–23], the determination of material trajectories from clinical quantitative computed tomography (QCT) scans remains an open question. Several recent preliminary studies have attempted to determine these trajectories on pieces of cortical femurs or vertebra [24,25]. Instead of the empirical HU-elastic property relations, we investigate herein inhomogeneous orthotropic properties derived by a micromechanical (MM) homogenization approach based on the microstructure and vascular porosity as suggested by Fritsch and Hellmich in Ref. [26] in combination with experimental data obtained at the osteon level [27]. This approach is chosen because the trabecular fabric cannot be determined using clinical CT scans and trabecular bone morphology cannot be determined (see Refs. [28–30] for the use of the fabric tensor in the case of  $\mu$ CT or high-resolution peripheral QCT may be applied). The MM-homogenization approach adopted is aimed at overcoming this limitation. A variation of the method presented herein was proven beneficial

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Table 1 Summary of the fresh-frozen femurs and CT scan resolution

Notation	Age	Gender	Side	Death reason	Slice thickness (mm)	Pixel (mm)
FF1	30	Male	Left	Car accident	0.75	0.78125
FF2	20	Female	Right	Stroke	1.5	0.72851
FF3	54	Female	Left	Cardiovascular disease	1.25	0.51757
FF4	63	Male	Right	Car accident	1.25	0.19531

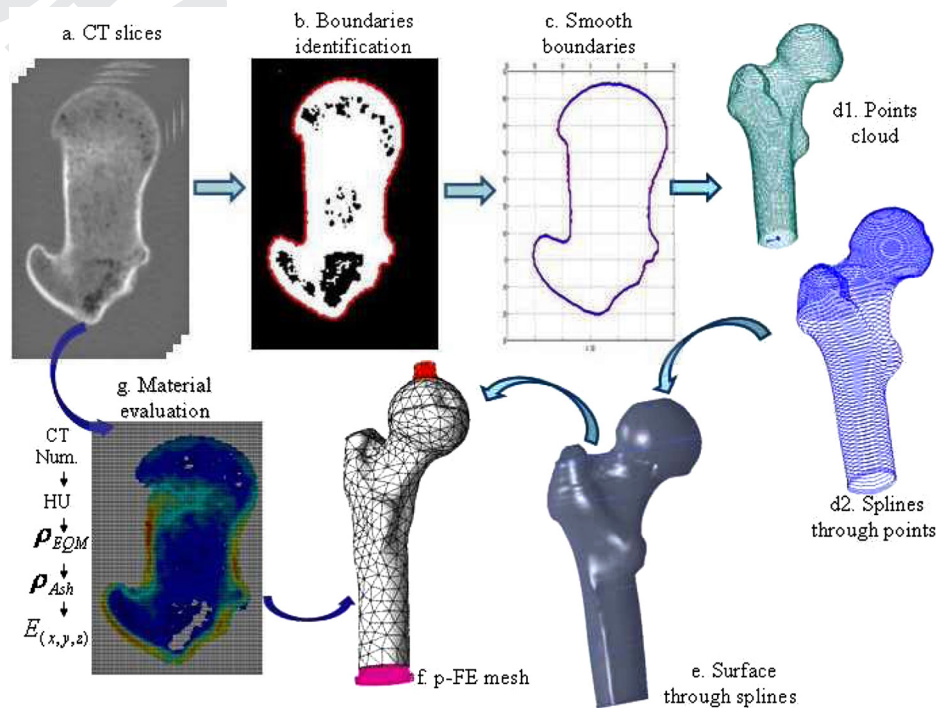


Fig. 1 Schematic flow chart describing the generation of the p-FE model from QCT scans: (a) typical CT-slice, (b) contour identification, (c) smoothing boundary points, (d1) points cloud representing the bone surface, (d2) close splines for all slices, (e) bone surface, (f) p-FE mesh, and (g) material evaluation from CT data

61 in the case of a human mandible simulation [31] and in Ref. [32]  
 62 it was used to generate a FE model of the femur with *inhomogeneous isotropic*  
 63 material properties, considering for the first time an  
 64 inhomogeneous Poisson ratio. Herein we extend the MM  
 65 approach by assigning inhomogeneous orthotropic material properties  
 66 to high-order FE models using QCT performed on four  
 67 femurs. In addition to determining the “bone matrix” material tensor  
 68 from experimental observations, we also investigate two methods  
 69 for determining material trajectories: (a) following the contour of  
 70 femur’s outer surface and (b) along the principal strain directions.  
 71 The FE results are compared to these obtained with isotropic  
 72 material and to the experimental observations on four different  
 73 fresh frozen femurs.

## 74 2 Methods

75 Four fresh-frozen human femurs were CT scanned, followed by  
 76 in vitro mechanical experiments. The QCT scans were manipulated  
 77 to generate patient-specific high-order FE bone models that  
 78 mimic the experimental conditions. The semiautomatic 3D reconstruction  
 79 of the femur’s geometry and generation of FE-meshes are detailed in Refs. [3,4,10]  
 80 and briefly summarized herein. All DICoM (Digital Imaging and Communication in  
 81 Medicine) format QCT scans were automatically manipulated by in-house computer  
 82 codes. First the scans are transformed into binary images in which nonzero pixels  
 83 belong to the femur domain and 0-value is

85 assigned to pixels associated with the background (scan resolutions  
 86 are summarized in Table 1). Exterior, interface (between trabecular and cortical  
 87 bone), and interior boundaries are traced in each slice and the points on these  
 88 boundaries are manipulated by a 3D smoothing algorithm to generate “smooth  
 89 boundary representation.” The same boundaries are used to determine the cortical,  
 90 trabecular, and cavity regions within the femur. An automeshing is thereafter  
 91 applied within the p-FE code STRESSCHECK that generates tetrahedral high-order  
 92 elements.<sup>1</sup> The entire algorithm (QCT to FE) is schematically illustrated in Fig. 1.

93 The polynomial degree over the elements was increased until convergence  
 94 in energy norm and strains at the points of interest was observed. Each FE model  
 95 consists between 3500 to 4500 elements [~150,000 degrees of freedom (DOFs)  
 96 at  $p=4$  and ~300,000 DOFs at  $p=5$ ]. The proper material properties are  
 97 directly extracted from the QCT scanned and assigned to each integration point  
 98 (512 points for each tetrahedral element) (details are provided in Sec. 2.1).  
 99 For an orthotropic material the material directions are also determined (as  
 100 explained in Sec. 2.1.3). Clamped boundary conditions are assigned to the  
 101 distal part and traction is applied on the head to mimic the experimental  
 102 conditions. The surface of the femur is accurately represented in the FE  
 103 model by using the blending function method [33]. In the p-  
 104  
 105  
 106  
 107

<sup>1</sup>STRESSCHECK is trademark of Engineering Software Research and Development, Inc, St. Louis, MO, USA.

108 version of the FE method convergence is realized by keeping a  
 109 fixed mesh (with relatively large elements) and increasing the  
 110 polynomial degree of the approximated solution. Therefore, the  
 111 accurate geometrical description of the domain must be realized  
 112 which is being accomplished by the use of blending-function  
 113 mapping.

114 **2.1 Material Properties Assignment.** Inhomogeneous ortho-  
 115 tropic properties derived based on the micromechanics analysis are  
 116 assigned to the FE models representing the femurs. Herein we com-  
 117 pare the mechanical response predicted by these models to FE mod-  
 118 els having isotropic material properties determined empirically.

119 *2.1.1 Isotropic Material Properties Determined by Empirical*  
 120 *Correlation.* Many empirical relations between Young’s modulus  
 121 and bone density, with a constant Poisson’s ratio are available,  
 122 see, e.g., Refs. [34–38]. In Refs. [3,4] we found that *p*-FE analyses  
 123 with the Keyak relationship [36] provide the closest results to in  
 124 vitro experiments on the proximal femur:

$$\rho_{EQM} = 10^{-3}(a \times HU - b) \left( \text{g/cm}^3 \right) \quad (1)$$

$$\rho_{Ash} = (1.22 \times \rho_{EQM} + 0.0523) \left( \text{g/cm}^3 \right) \quad (2)$$

$$E_{Cort} = 10200 \times \rho_{Ash}^{2.01} \text{ (MPa)} \quad (3)$$

$$E_{Trab} = 5307 \times \rho_{Ash} + 469 \text{ (MPa)} \quad (4)$$

125 where  $\rho_{EQM}$  is the equivalent mineral density;  $\rho_{Ash}$  is the ash den-  
 126 sity;  $E_{Cort}$ ,  $E_{Trab}$  are the Young’s moduli in the cortical and tra-  
 127 bular regions and the parameters  $a$  and  $b$  are determined by the  
 128  $K_2HPO_4$  phantoms in the CT-scan. Poisson’s ratio is constant  
 129  $\nu = 0.3$ .

130 *2.1.2 MM-based Orthotropic Material Properties.* A contin-  
 131 uum MM-based model (details are available in Refs. [31,32] is  
 132 applied on the QCT scans to determine (nonempirical) relations  
 133 between orthotropic elasticity tensor components and HU. It is  
 134 based on two consecutive steps [26,31]:

- Based on voxel average rules for the attenuation coeffi-  
 135 cients, we assign to each voxel the volume fraction occupied  
 136 by water (marrow) and that occupied by solid bone matrix.  
 137 The volume fraction is identical to the vascular porosity, as  
 138 given by

$$\phi(x) = \begin{cases} \frac{HU_{BM} - HU(x)}{HU_{BM}} & \forall HU \leq 1600 \\ 0 & \text{otherwise} \end{cases} \quad (5)$$

139 We denote by  $x$  the position of the individual voxel with  
 140  $HU = 0$  representing pure water, and  $HU_{BM} \geq 1600$  repre-  
 141 sents a “perfectly compact” bone in a human femur (see Ref.  
 142 [32] for details). The lower  $HU$  values refer to very porous  
 143 trabecular bone, with a vascular porosity ( $\phi$ ) close to 100%.  
 144 At the upper end of the  $HU$ , values are identified as associ-  
 145 ated with vanishing vascular porosity  $\phi \sim 0$ .

- By means of a MM model for bone based on mechanical  
 146 properties of solid bone matrix and of water, we convert  $\phi$   
 147 into voxel-specific orthotropic material tensor components.  
 148 The model, cast in the framework of random homogenization  
 149 theory [31], is of the Mori–Tanaka type, so that the effective  
 150 stiffness tensor  $C_{eff}$  of the bone at position  $x$  is given by

$$C_{eff} = \{ \phi C_{H_2O} : [\mathbb{I} + \mathbb{P}_{cyl} : (C_{H_2O} - C_{BM})]^{-1} + (1 - \phi) C_{BM} \} : \{ \phi [\mathbb{I} + \mathbb{P}_{cyl} : (C_{H_2O} - C_{BM})]^{-1} + (1 - \phi) \mathbb{I} \}^{-1} \quad (6)$$

151 where  $\mathbb{I}$  is the fourth-order identity tensor (A3),  $\mathbb{P}_{cyl}$  is the  
 152 fourth-order Hill tensor (A4) accounting for the cylindrical  
 pore shape in a bone matrix of stiffness  $C_{BM}$  (A1),  $C_{H_2O}$  (A2)

is the bulk elastic stiffness and colons denote the second-order  
 tensor contraction (see further details in Appendix A). 153

The bone-matrix material tensor can be obtained at the microlevel 154  
 by either ultrasonic [40] or nanoindentation techniques [27]. The 155  
 identification of orthotropic constant for bone is still an open 156  
 problem therefore six relevant studies are presented in Table 4 in 157  
 Appendix A to provide an overview of the available knowledge 158  
 and to demonstrate the material properties range. A preliminary 159  
 investigation was undertaken by which the properties in Refs. [40] 160  
 and [27] were applied in the MM model: 161

$$C_{BM}^{[I]} = \begin{pmatrix} 18.5 & 10.3 & 10.4 & 0 & 0 & 0 \\ 10.3 & 20.8 & 11.0 & 0 & 0 & 0 \\ 10.4 & 11.0 & 28.4 & 0 & 0 & 0 \\ 0 & 0 & 0 & 12.9 & 0 & 0 \\ 0 & 0 & 0 & 0 & 11.5 & 0 \\ 0 & 0 & 0 & 0 & 0 & 9.3 \end{pmatrix} \text{ (GPa)},$$

$$C_{BM}^{[11]} = \begin{pmatrix} 20.8 & 10.4 & 16.6 & 0 & 0 & 0 \\ 10.4 & 22.1 & 16.5 & 0 & 0 & 0 \\ 16.6 & 16.5 & 41.9 & 0 & 0 & 0 \\ 0 & 0 & 0 & 15.4 & 0 & 0 \\ 0 & 0 & 0 & 0 & 11.2 & 0 \\ 0 & 0 & 0 & 0 & 0 & 9.4 \end{pmatrix} \text{ (GPa)} \quad (7)$$

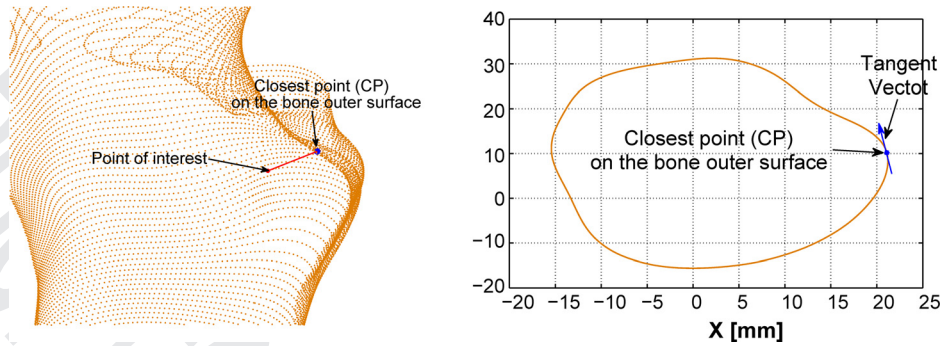
The resulting inhomogeneous orthotropic material properties were 162  
 assigned on our FE models. Those studies were chosen since they 163  
 both reported on all required material constant, they used different 164  
 experimental tool and they represent well the range of the reported 165  
 material properties. Our models shows that  $C_{BM}^{[I]}$  results in a 166  
 “weak” response compared to several experimental results [40]. 167  
 Based on our preliminary results the current study adopts the elas- 168  
 tic constant suggested in Ref. [27] to be used in the MM model in 169  
 conjunction with the *p*-FE model. 170

171 *2.1.3 Determination of Material Trajectories.* For an ortho-  
 172 tropic material the material trajectories at each point have to be  
 173 determined. The (vectorial) material trajectories cannot be deter-  
 174 mined clearly from a clinical CT scan thus additional information  
 175 like the characteristic density distributions within the bony organ  
 176 or the organ’s surface description may be used [26,41]. Herein,  
 177 we tested two alternative assumptions to obtain the inhomogene-  
 178 ous (voxel-specific) material trajectories: (a) either following the  
 179 femur’s outer geometry (determined by the biological evolution  
 180 of the bone to best carry the loading) or (b) following the principal  
 181 strains (see, e.g., Ref. [42]). In both cases, we consider the longi-  
 182 tudinal direction (axis-3) as the “stiff” material direction (having  
 183 the largest Young’s modulus) and the other two transverse direc-  
 184 tions as weak (having the smaller Young’s modulus). The trans-  
 185 verse plane is rather isotropic ( $E_1$  is close in value to  $E_2$ ). In this  
 186 section we outline the algorithms for determining the normal vec-  
 187 tors that represent material trajectories.

188 *2.1.3.1 Material trajectories according to outer surface*  
 189 *geometry.* By assuming that material trajectories follow the outer  
 190 surface, we developed a three-step algorithm to determine these:

- 191 Creation of smooth boundary surfaces from CT data: CT  
 192 data is processed as shown in Figs. 1(b), 1(c), and 1(d1):  
 193 Bone borders are detected at each CT slice, then a 3D  
 194 smoothing algorithm is applied and finally a cloud of  
 195 point is kept that represent femur’s boundary used for  
 196 surface reconstruction.
- 197 Computation of the closest point (CP) on femurs outer  
 198 surface: For each point of interest (POI) at which trajecto-  
 199 ries are sought, the closest point (CP) on the outer surface





**Fig. 2** Left closest point (CP) on the bone outer surface to a specific POI. Right: tangent (circumferential) direction.

200 is identified, by minimizing the distance among all points  
201 on the surface:

$$R = \min_{x \text{ on surface}} \times \left[ \sqrt{(x_{\text{surface}} - x_{\text{POI}})^2 + (y_{\text{surface}} - y_{\text{POI}})^2 + (z_{\text{surface}} - z_{\text{POI}})^2} \right] \quad (8)$$

202 see Fig. 2, left. The material trajectories related to the CP  
203 are then assigned to the POI.

204 c) Computing radial, circumferential (tangent), and longitudinal  
205 trajectories: The circumferential (tangent) direction  
206 is the vector that follows the direction of a spline at the  
207 CP, see Fig. 2, right. The longitudinal direction is the one  
208 parallel to the bone surface and associated to the gradient  
209 along the  $z$  axis. The closest points to the CP on the spline  
210 below and above the spline on which the CP is located  
211 are found. The vector connecting these two points is the  
212 longitudinal direction. The radial (normal) direction is  
213 easily computed by the vector product of the longitudinal  
214 and tangent vectors. The three normalized direction vectors  
form the material trajectories at the POI.

215 **2.1.3.2 Material trajectories following principal strains.**  
216 Another possibility to determine material trajectories is motivated  
217 by Wolff's law [43], and associated with strains principal direc-  
218 tions. The main assumption is that bone tissue orientation correlates  
219 well to principal strain direction [44]. The stance position loading  
220 (the magnitude of load does not influence the principal directions  
221 but only the magnitude of the strains) is used for approximating the  
222 principal strain directions assuming a homogeneous isotropic material.  
223 This approach is also motivated by the recent experimental  
224 results in Ref. [45], which show that the principal strain directions  
225 on the femur's surface is almost independent on the loading conditions  
226 on the femur's head, covering the range of directions spanned  
227 by the hip joint force. Although the principal strains magnitude varied  
228 greatly between loading configurations, the principal strain  
229 direction varied very little. This suggests that the anatomy and the  
230 distribution of anisotropic material properties in the proximal femur  
231 is probably not strongly affected by the various loading directions,  
232 and can be determined from the direction of the principal strains  
233 resulting from the stance loading configuration.

234 At each POI the principal strains are computed and the  
235 "stiffest" direction 3 is associated with the largest absolute value  
236 of the strain tensor eigenvalues. For each POI at  $x$ , the coordinate  
237 system along principal strain directions is denoted by  $X_\alpha^m$ ,  
238  $\alpha = 1, 2, 3$  (3 is the "stiffest"). Since the material tensor  $C^m$   
239 is assumed to be along  $X_\alpha^m$  it is transformed into the femur's coordi-  
240 nate system by [46]

$$C_{ijkl} = C_{\alpha\beta\gamma\delta}^m \ell_{\alpha i} \ell_{\beta j} \ell_{\gamma k} \ell_{\delta l} \quad (9)$$

241 where  $\ell_{\alpha i} \equiv \cos(X_\alpha^m, x_i)$

242 **2.1.4 The "Stiffest" Material Trajectory.** We preformed numerical  
243 tests to evaluate the differences of obtained trajectories by  
244 the two proposed methods. Figure 3 presents the difference in the  
245 material trajectories in the "stiffest" on a typical 2-3 plane. Several  
246 points of interest are presented to exemplify the specific distribu-  
247 tion of the "stiffest" material trajectory.

248 In most of the shaft region the "stiffest" trajectory is orientated  
249 along the femur's  $x_3$  axis and is in agreement with other studies  
250 [25]. For POIs close to femur's outer boundary in the head and  
251 neck regions, both methods result in similar directions. In the central  
252 femoral's head and neck; on the other hand, the directions are  
253 different.

254 The different methods' influence on the FE results was numerically  
255 tested. Same MM orthotropic law but with material trajectories  
256 based on bone's geometry and the other based on principal  
257 strains were applied to the FE models. Strain and displacements  
258 on bone surface were compared. No significant differences in the  
259 FE result was demonstrated (mean error was less than 5%). Since  
260 there is no clear advantage of one method over the other, we  
261 determine the material trajectories following the principal strains.  
262 The main reasons for this choice are the complexity and long  
263 computational times when using the geometric based method  
264 compared to the robustness in the principal strains method.

265 **2.2 Mechanical Experiments.** In vitro experiments on four  
266 fresh-frozen proximal femurs summarized in Table 1 were used to  
267 assess the validity of FE simulations.

268 The experiments simulate a simple stance position configura-  
269 tion in which the femurs were loaded through their head while  
270 inclined at two different inclination angles ( $0^\circ$  and  $20^\circ$ ) as shown  
271 in Fig. 4. We measured the vertical and horizontal displacements  
272 of femur's head, the strains at the inferior and superior parts of the  
273 neck, and on the medial and lateral femoral shaft. Between 5 and  
274 10 strain-gauges were bonded on each of the tested femurs. In all  
275 experiments a linear response between force and displacements  
276 and strains was observed beyond 200N preload. The experimental  
277 error is within a  $\pm 5\%$  range. This range was estimated by the  
278 measurements error (calibration to known displacements/loads/  
279 strains), deviation between consecutive measurements and estima-  
280 tion of the linear response (details are provided in Refs. [4,10]).

### 3 Results 281

282 **3.1 Comparison of Material Properties.** The MM-based  
283 orthotropic constants are compared to the empirical isotropic  
284 Young's modulus given by Refs. [34,47] in Fig. 5 [empirical shear  
285 modulus is computed with ( $\nu = 0.3$ )]. In the longitudinal direc-  
286 tion ( $E_3$ ) is higher for  $HU < 1400$  while in the transverse detec-  
287 tions ( $E_1$  and  $E_2$ ) are smaller compared to the isotropic empirical  
288  $E$ , whereas the Poisson ratio is clearly nonconstant. Similar shear  
289 moduli are obtained for  $HU < 600$ . One can notices that  
290 ( $0.9 < E_1/E_2 < 1$ ) and ( $0.3 < E_2/E_3 < 0.6$ ), supporting our

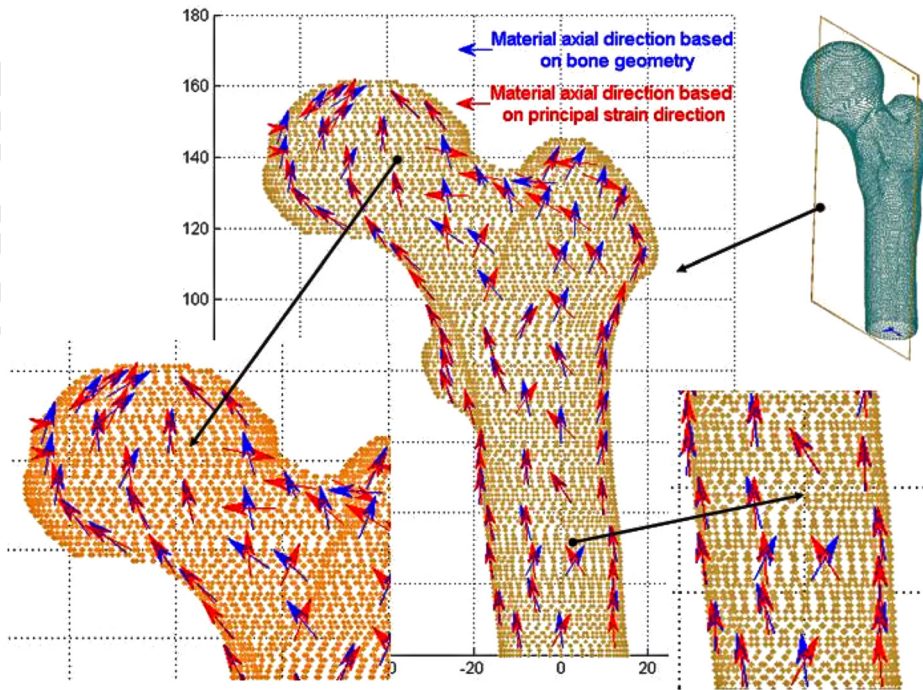


Fig. 3 Axial material trajectories throughout a femur, zoomed portion in the trabecular area and femoral neck (left) and zoomed portion in cortical area and shaft (right)

291 observation that the longitudinal direction is the “stiff” one and  
292 that the transverse direction is rather isotropic.

293 **3.2 FE Results Using Isotropic and Orthotropic Properties**  
294 **Compared to Experimental Observations.** We monitor the p-  
295 FE convergence in energy norm, obtaining for  $p = 4$  an error  
296 smaller than 2%. The experimental measured strains on femur’s  
297 surface at several locations and the vertical and horizontal displacements  
298 of the femur’s head are used for comparison purposes. At the  
299 same locations the strains in the direction of the strain-gauges and  
300 displacements were extracted from the FE solutions. The FE strains  
301 are reported as the average over a face element on which the strain-  
302 gauge is bonded in reality because the strain-gauge (SG) readings

are also an averaged value over SG’s length. In Fig. 6 we present, 303  
for example, these locations on the FE model of FF3. To estimate 304  
quantitatively the validity of the various FE models, we compared 305  
*all measured data* (both displacements and strains for the four 306  
femurs of interest) to the computed data, i.e., we computed the 307  
linear regression between the data sets and the normalized root mean 308  
squared error (NRMSE). Figure 7 presents a linear regression of 309  
the experimental results compared to FE predictions, for both dis- 310  
placements and strains (for a load of 1000N). These linear regres- 311  
sions demonstrate the general quality assessment of the analysis. A 312  
very good correlation is obtained between the predicted and meas- 313  
ured strains and displacements for the empirical based isotropic 314  
model. The orthotropic model makes the model “less stiff” but still 315  
a good correlation to the experiments is observed. In spite of 316

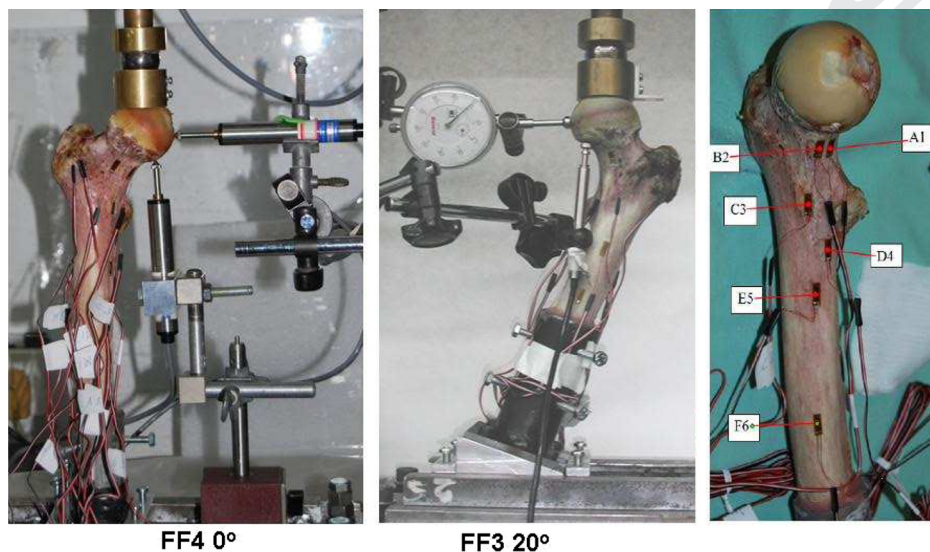


Fig. 4 Typical experiments on fresh-frozen bones (FF3 and FF4) at different inclination angles. Right: Strain gauges location at the neck and shaft regions.



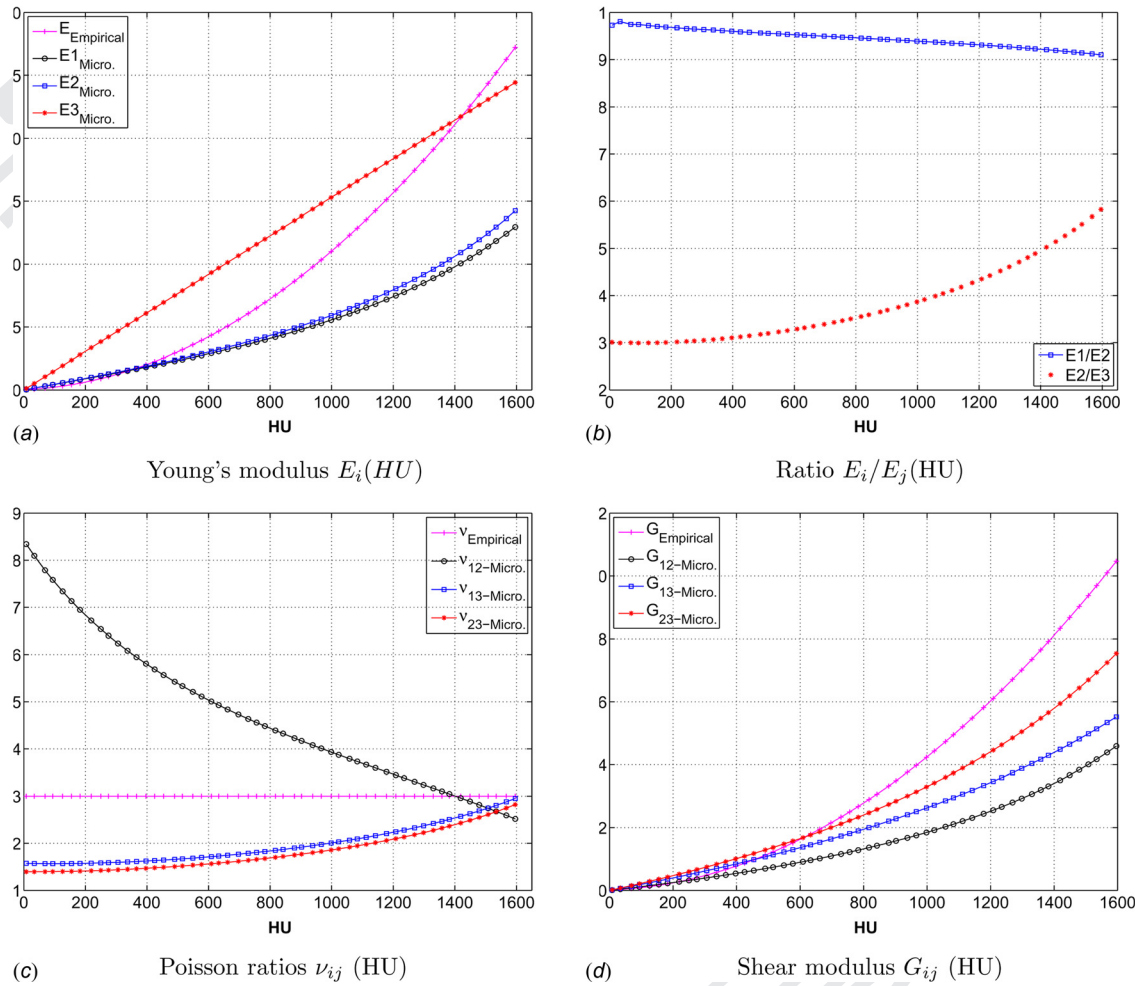


Fig. 5 (a)  $E(HU)$  relation for orthotropic MM-based and isotropic empirical-based models [34,47]. (b) Ratio of Young's modulus in different directions. (c) Poisson ratio dependence on HU for MM-based model ( $\nu = 0.3$  for empirically based model). (d) Shear moduli relation to HU for MM-based and empirically based models.

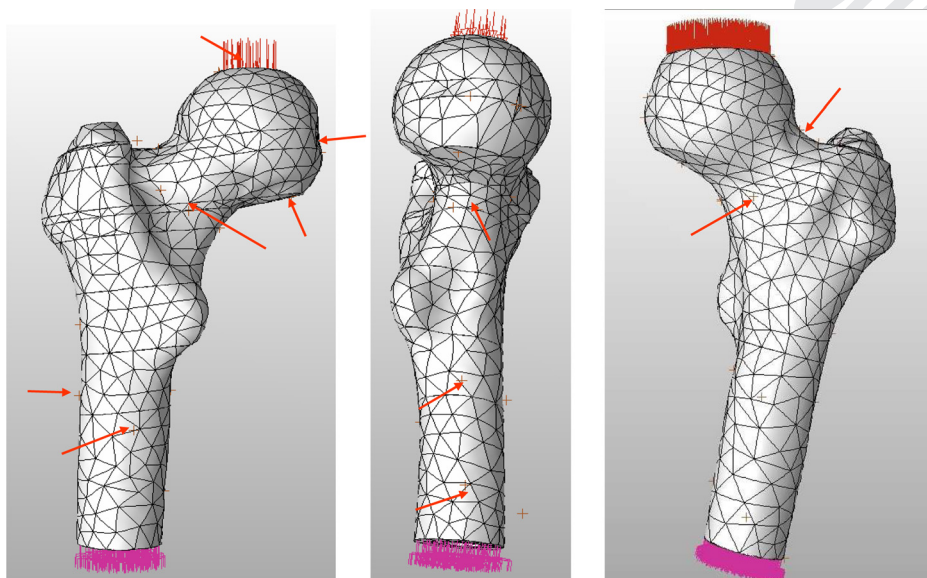


Fig. 6 FF3-FE model and locations at which displacements and strains were computed and measured in the experiment

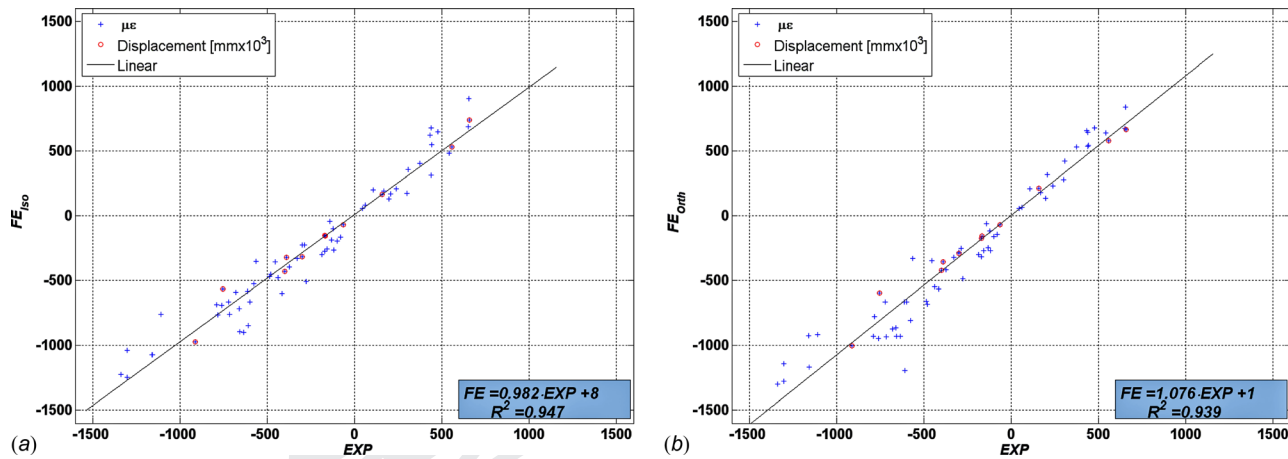


Fig. 7 Comparison of the computed strains + and displacements ⊕ to the experimental observations. Material properties assigned by two different strategies in the FE models: (a) empirical-based, (b) MM-based.

317 differences presented in Fig. 7, for biological structures the results  
 318 are of comparable accuracy (less than 10%).

319 Since the loading in our experiments implies mostly normal  
 320 strains and displacements in the longitudinal direction with no  
 321 shearing or torsion, the close results between isotropic and MM  
 322 orthotropic based models are very reasonable. To further examine  
 323 the differences between the two models, we compare separately  
 324 the results obtained for different inclination angles (0° and 20°).  
 325 In addition, the results were divided into three main groups of in-  
 326 terest: strains at the femur’s neck region, strains at the femur’s  
 327 shaft region and displacements of femur’s head. A summary of  
 328 the comparisons is presented in Table 2. To quantify and empha-  
 329 size the differences between the isotropic versus the orthotropic  
 330 model, a comparison between the FE results based on the different  
 331 material models is presented in Table 3.

332 **4 Discussion**

333 It is widely accepted that bone-tissue is orthotropic rather than  
 334 isotropic, and that the main orientation of bone’s microstructure  
 335 appears to adapt along principal strains [20,42]. Studies compar-  
 336 ing the mechanical response under isotropic and very restricted  
 337 orthotropic material properties under a stance loading condition,

Table 2 Summary of linear regression and NRMSE: FE results compared to experimental observations. All results = both strains and displacements, SGs: strain gauge results, NRMSE

$$= \sqrt{\frac{\sum_{i=1}^n (x_i - y_i)^2}{n} / [\max(x) - \min(x)]}.$$

Region and tilt	Material model	Slope	R <sup>2</sup>	NRMSE
All results	iso.	1.01	0.96	0.06
Tilt 0°	orth.	1.09	0.96	0.07
All results	iso.	0.88	0.91	0.08
Tilt 20°	orth.	1.04	0.88	0.10
All results	iso.	0.98	0.95	0.06
Tilt 0°, 20°	orth.	1.08	0.94	0.07
All SGs	iso.	1.02	0.96	0.06
Tilt 0°	orth.	1.10	0.96	0.08
All SGs	iso.	0.87	0.90	0.09
Tilt 20°	orth.	1.03	0.87	0.11
All SGs	iso.	0.98	0.94	0.06
Tilt 0°, 20°	orth.	1.09	0.93	0.08
Displacements	iso.	0.99	0.98	0.04
Tilt 0°, 20°	orth.	1.0	0.99	0.04
Shaft region	iso.	1.1	0.94	0.07
Tilt 0°, 20°	orth.	1.06	0.93	0.08
Neck region	iso.	0.88	0.96	0.06
Tilt 0°, 20°	orth.	1.09	0.94	0.09

conclude that a very similar response is obtained; however, none  
 of these used models that represent realistic local anisotropy  
 throughout the entire bone and no comparison to experimental ob-  
 servation was reported [13,17,18]. Encouraged by promising stud-  
 ies addressing anisotropy in bones [31,32] we developed herein a  
 systematic micromechanics-based algorithm to evaluate inhomog-  
 eneous orthotropic bone’s properties based on QCT scans. Mate-  
 rial parameters obtained by the MM approach are shown to lie  
 within the range of these obtained by empirical methods in vari-  
 ous past publications.

Two automatic methods to determine the anisotropic material  
 trajectories were found to be very similar in the shaft and close to  
 bone’s surface in the head, but differ in the internal portion of the  
 head. The MM based material properties in femur’s head were  
 compared to experimental observations in a recent study by  
 Ohman et al. [48], adding another level of confidence in the pro-  
 posed MM algorithm. Ohman et al. extracted cylindrical speci-  
 mens from femoral heads with alignment and misalignment to the  
 trabecular main direction of approximately 20°. The Young’s  
 modulus values reported in Ref. [48] were 2.76(±1.06) GPa for  
 the “aligned” and 1.59(±0.66) GPa for the “misaligned” groups,  
 related to 100 < HU < 250. About 40% difference was found in  
 measured Young’s modulus in the ‘misaligned’ specimens, lower  
 than these in the “aligned” ones. In all femurs investigated herein  
 we found that the Young’s modulus is in the same range as  
 reported by Ohman et al. and varies about 30–40% if rotated by  
 20°, as observed in Ref. [48].

QCT-scans were used to generate isotropic and orthotropic FE  
 models of four fresh-frozen human femurs harvested from both  
 genders and age span of 20–63 years-old. The reliability of the FE  
 results and the influence of the assigned material properties was  
 estimated by comparison to strains and displacements obtained by  
 in vitro experiments. It is important to realize that the mechanical  
 tests represent a very simple loading condition which is not neces-  
 sary physiological. The simple loading condition is due to experi-  
 mental constraints and aimed at reducing the complexity of the  
 mechanical response.

Table 3 Linear regression and NRMSE: isotropic versus ortho-  
 thotropic FE results (tilt 0° and 20°)

Region of interest	Slope	R <sup>2</sup>	NRMSE
All results	1.08	0.97	0.05
All SGs	1.09	0.97	0.06
Displacements	1.01	0.99	0.02
Shaft region	0.97	0.99	0.02
Neck region	1.24	0.97	0.10

375 The strains and displacements computed by the  $p$ -FEMs with in-  
 376 homogeneous MM-based orthotropic properties in conjunction  
 377 with material trajectories along principal strains correlate well with  
 378 the experimental observations. The comparison presented in Table  
 379 II demonstrates that if the principal strains or head displacements  
 380 are of interest, the differences between the results of the isotropic  
 381 empirical based model and the orthotropic MM model are small  
 382 and in the range of the experimental errors. On the other hand,  
 383 a more pronounced difference is noticed between the isotropic and  
 384 orthotropic FE results when only strains in the femoral neck are  
 385 compared. To assess if the orthotropic and isotropic models yield  
 386 similar results for a more complex state of loading we applied on  
 387 the same FE models a compression load on femur's head in addition  
 388 to a torsional load (resulting in a moment along the  $z$  axis).  
 389 Because experiments with complex loading configurations on  
 390 fresh-frozen human femurs are unavailable at this time, we only  
 391 address the numerical results. Comparing the FE predicted strains  
 392 by the different material models show a significant differences  
 393 between them (slope = 1.35,  $R^2 = 0.96$ , NRMSE = 0.14, and  
 394 slope = 1.19,  $R^2 = 0.99$ , NRMSE = 0.09 for displacements). This  
 395 numerical study demonstrates that when a more complex loading  
 396 condition is applied, the FE predicted mechanical response of the  
 397 femur is considerably different (the orthotropic model is signifi-  
 398 cantly less stiff). Nevertheless, since no experimental observations  
 399 for such loading is available, it is impossible to assess at this time  
 400 which of the two models better represents the reality.

401 The scant quantitative studies that identify the difference in the  
 402 mechanical response when isotropic viz. orthotropic materials are  
 403 considered, seem to suggest that the difference is negligible  
 404 [13,17]. This conclusion, in view of our results herein, seems to  
 405 be restricted to simplified loading conditions and not accurate for  
 406 all regions of interest. Since the anisotropy is more pronounced in  
 407 the femoral neck region this area is more sensitive to the different  
 408 material assignment strategy. Similar conclusions (showing a differ-  
 409 ence in the mechanical response if isotropic or orthotropic  
 410 materials were applied) were obtained in a recent work on a differ-  
 411 ent organ (the mandible) [31]. In an ongoing research on  $p$ -FE  
 412 predictions of failure initiation in the femur, a considerable differ-  
 413 ence is noticed between failure loads predicted by the isotropic  
 414 and orthotropic models [49].

415 Several limitations of the present work have to be discussed. (a)  
 416 Only four cadaver proximal femurs were investigated. (b) Only  
 417 simple loading conditions were considered for the validation pro-  
 418 cess. (c) The orthotropic MM model is based on a clinical CT  
 419 scan and does not accurately represent the trabecular bone mor-  
 420 phology. (d) The differences between the isotropic model and the  
 421 orthotropic MM model are in the range of the experimental errors  
 422 so unequivocal conclusions cannot be drawn regarding which  
 423 model is better. Both models seem to provide very good correla-  
 424 tion to experimental observations under the simplified loading  
 425 condition.

426 To conclude, this study demonstrates the ability to apply  $p$ -FE  
 427 technology to analyze patient-specific femurs with inhomogeneous  
 428 micromechanics-based orthotropic material properties with  
 429 material trajectories along the principal strains. The methods were  
 430 numerically verified and validated by experimental observations.  
 431 Future in vitro experiments resulting in a more complex state of  
 432 stresses in the femur (a more realistic physiological load) is  
 433 planned to further corroborate our conclusions. In this future study  
 434 we will examine if indeed the inhomogeneous orthotropic material  
 435 properties are necessary (as opposed to a simplified isotropic  
 436 assumption) for a reliable simulation of the mechanical response,  
 437 especially in the head and neck locations of the proximal femur.  
 438 This is of major importance to failure analysis of osteoporotic  
 439 bones, and the possibility to predict such failures as a function of  
 440 bones' density and local geometry. We also intend to investigate  
 441 "more physiological" loadings that may have a large influence on  
 442 femur's mechanical response [45] by considering  $p$ -FE models  
 443 loaded according to Refs. [47,50,51] and results compared to ex-  
 444 vivo experiments.

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**Appendix A: Micromechanics-Based Inhomogeneous Orthotropic Material Properties**

Several of the micromechanics-based relations required for the  
 computation of  $\mathbb{C}_{\text{eff}}$  are provided herein following Refs. [26,31].  
 The orthotropic material tensor  $\mathbb{C}_{\text{eff}}$  is represented in matrix nota-  
 tion, by replacing the subscripts  $ij$  (or  $kl$ ) by  $m$  (or  $n$ ) using the  
 Voigt notation:

$$\begin{matrix} ij \text{ (or } kl) & 11 & 22 & 33 & 23 \text{ or } 32 & 31 \text{ or } 13 & 21 \text{ or } 12 \\ m \text{ (or } n) & 1 & 2 & 3 & 4 & 5 & 6 \end{matrix} \quad \mathbb{C}_{\text{BM}}^{\text{exp}} = \begin{pmatrix} C_{1111} & C_{1122} & C_{1133} & 0 & 0 & 0 \\ C_{1122} & C_{2222} & C_{2233} & 0 & 0 & 0 \\ C_{1133} & C_{2233} & C_{3333} & 0 & 0 & 0 \\ 0 & 0 & 0 & 2C_{2323} & 0 & 0 \\ 0 & 0 & 0 & 0 & 2C_{1313} & 0 \\ 0 & 0 & 0 & 0 & 0 & 2C_{1212} \end{pmatrix} \quad (\text{A1})$$

and the following relations hold:

$$C_{H_2O} = 3 \cdot k_{H_2O} J \text{ where } J_{ijkl} = \frac{1}{3} \delta_{ij} \delta_{kl} \text{ and } k_{H_2O} = 2.3 \text{ (GPa)} \quad (\text{A2})$$

$$I_{ijkl} = (\delta_{ik} \delta_{jl} + \delta_{il} \delta_{jk}) / 2 \quad (\text{A3})$$

$$P_{\text{cyl}} = \frac{1}{2\pi} \int_0^{2\pi} \mathbb{B} d\Phi \quad (\text{A4})$$

$$B_{ijkl} = \frac{1}{4} (\xi_i \bar{G}_{jk} \xi_l + \xi_j \bar{G}_{ik} \xi_l + \xi_i \bar{G}_{jl} \xi_k + \xi_j \bar{G}_{il} \xi_k) \quad (\text{A5})$$

$$\mathbf{G} = \mathbf{K}^{-1} \quad (\text{A6})$$

$$\mathbf{K} = \xi \cdot \mathbb{C}_{\text{BM}} \cdot \xi, \quad \mathbf{K}_{jk} = \xi_i \mathbb{C}_{\text{BM},ijkl} \xi_l \quad (\text{A7})$$

$$\xi = \cos \Phi \hat{e}_1 + \sin \Phi \hat{e}_2 + \hat{e}_3 \quad (\text{A8})$$

where center dots denote the first-order tensor contraction (also  
 called inner product).

The compliance matrix  $\mathbb{S} = \mathbb{C}^{-1}$  for orthotropic materials  
 presented in Eq. (A9) may be represented by the nine material  
 constants ( $E_1, E_2, E_3, G_{12}, G_{13}, G_{23}, \nu_{12}, \nu_{13}, \nu_{23}$ ), where 1 refers  
 to the radial, 2 refers to the circumferential, and 3 refers to the  
 longitudinal (axial) direction of the bone material:

$$\mathbb{S} = \begin{pmatrix} \frac{1}{E_1} & \frac{-\nu_{21}}{E_2} & \frac{-\nu_{31}}{E_3} & 0 & 0 & 0 \\ \frac{-\nu_{12}}{E_1} & \frac{1}{E_2} & \frac{-\nu_{32}}{E_3} & 0 & 0 & 0 \\ \frac{-\nu_{13}}{E_1} & \frac{-\nu_{23}}{E_2} & \frac{1}{E_3} & 0 & 0 & 0 \\ 0 & 0 & 0 & \frac{1}{G_{23}} & 0 & 0 \\ 0 & 0 & 0 & 0 & \frac{1}{G_{31}} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{1}{G_{12}} \end{pmatrix} \quad (\text{A9})$$

Bone matrix coefficients are computed by the nine ortho-  
 tropic elastic properties see Ref. [52]. Six relevant studies are pre-  
 sented in Table 4 to provide an overview of the available current  
 knowledge and to demonstrate the material properties range.



**Table 4 Orthotropic elastic constant for bone matrix. Young's and Shear modulus are in GPa, SD = standard deviations, NR = not reported. 1: radial, 2: circumferential, and 3: longitudinal (axial) direction of the bone material.**

Ref. year	Bone Site	Exp. method	No. of bones (specimen)	$E_1^*$ (SD)	$E_2$ (SD)	$E_3$ (SD)	$G_{12}$ (SD)	$G_{13}$ (SD)	$G_{23}$ (SD)	$\nu_{12}$ (SD)	$\nu_{13}$ (SD)	$\nu_{23}$ (SD)
[40] 1984	femur	ultr.	5 (60)	12 (1)	13.4 (1.1)	20 (1.5)	4.53 (0.3)	5.61 NR	6.23 NR	0.235 NR	0.371 NR	0.376 NR
[27]a 2009	femur	nano.	1 (22)	9.17 (0.63)	17.28 (1.89)	24.66 (2.71)	4.69 (0.37)	5.61 (0.47)	7.68 (0.53)	0.286 (0.024)	0.557 (0.022)	0.248 (0.012)
[53] 1996	tibia	ultr.	8 (96)	11.7 (1.3)	12.2 (1.4)	20.7 (1.9)	4.1 (0.5)	5.17 (0.6)	5.7 (0.5)	0.23 (0.035)	0.417 (0.048)	0.42 (0.074)
[54] 2002	tibia	nano.	1 (12)	16.6 (1.5)	17 (2.2)	25.1 (2.1)	NR	NR	NR	NR	NR	NR
[55] 2002	femur	nano.	9 NR	NR	NR	21.8 (2.1)	NR	NR	NR	NR	NR	NR
[56] 2008		MM based	—	16.4	18.7	22.8	7.2	7.1	8.4	0.25	0.33	0.33

\*In this case 1,2,3 represent the osteon level radial, circumferential and longitudinal directions.

**Appendix B: Empirical Relations for Inhomogeneous Orthotropic Material Properties**

(anterior-posterior), and 3 the axial (superior-inferior) directions. These connections are denoted by the subscript "Wirtz" in the following.

AQ4 477 This appendix presents the MM-based properties compared to 478 orthotropic material properties based on empirical correlation 479 with density provided in Refs. [20,21]:

Other empirical relations (for the cortical and trabecular bone) that correlate the Young's modules to the density  $\rho$  are reported in Ref. [58]:

$$E_{1/2}^{Cort} = 2314 \cdot \rho_{app}^{1.57} \text{ (MPa)} \quad (B1)$$

$$E_1^{Cort} = -6087 + 10 \cdot \rho \text{ (MPa)} \quad (B7)$$

$$E_3^{Cort} = 2065 \cdot \rho_{app}^{3.09} \text{ (MPa)} \quad (B2)$$

$$E_2^{Cort} = -4007 + 9 \cdot \rho \text{ (MPa)} \quad (B8)$$

$$E_{1/2}^{Trab} = 1157 \cdot \rho_{app}^{1.78} \text{ (MPa)} \quad (B3)$$

$$E_3^{Cort} = -6142 + 14 \cdot \rho \text{ (MPa)} \quad (B9)$$

$$E_3^{Trab} = 1904 \cdot \rho_{app}^{1.64} \text{ (MPa)} \quad (B4)$$

$$E_1^{Trab} = 0.004 \cdot \rho^{2.01} \text{ (MPa)} \quad (B10)$$

$$\nu_{ij} = \text{const} \quad (B5)$$

$$E_2^{Trab} = 0.01 \cdot \rho^{1.88} \text{ (MPa)} \quad (B11)$$

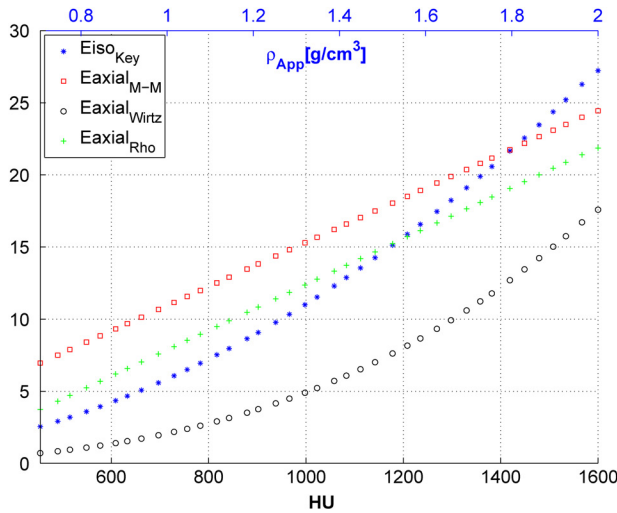
$$G_{ij} = \frac{G_{ij}^{max} \cdot \rho_{app}^2}{\rho_{max}^2} \text{ (MPa)} \quad (B6)$$

$$E_3^{Trab} = 0.58 \cdot \rho^{1.3} \text{ (MPa)} \quad (B12)$$

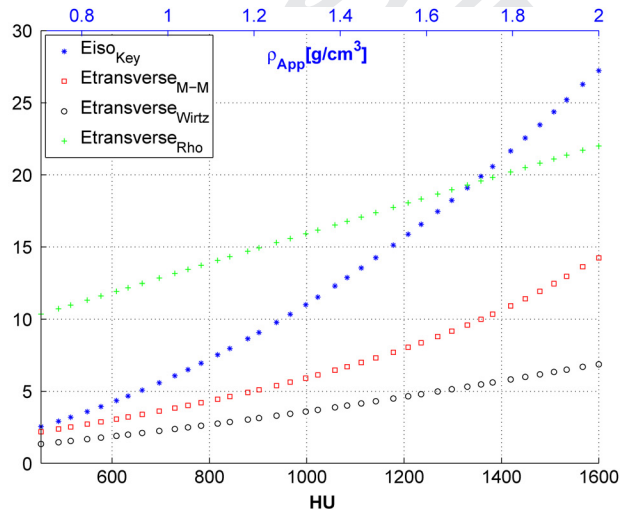
$$\nu_{ij} = \text{const} \quad (B13)$$

$$G_{ij} = f(E_i, E_j, \nu_{ij}) \text{ (MPa)} \quad (B14)$$

480 \* $\rho_{max}$  is the maximum apparent density and  $\rho_{app}$  is the apparent 481 density as measured by QCT according to Ref. [57]. Here 1 denotes 482 radial (medial-lateral in original manuscripts), 2 circumferential



(a) Longitudinal direction



(b) Transversal direction

**Fig. 8 Young's modulus in different directions versus HU or apparent density for cortical bone.  $E_{iso\_Key}$  represents the isotropic Young's modulus in Ref. [36],  $E_{axial\_MM}$  and  $E_{transverse\_MM}$  are the axial and the transversal Young's modulus computed by the micro-mechanics model,  $E_{axial\_Wirtz}$  and  $E_{transverse\_Wirtz}$  are the axial and transversal Young's modulus according to Eqs. (B2) and (B1), and  $E_{axial\_Rho}$  and  $E_{transverse\_Rho}$  is the axial and transversal Young's modulus according to Eqs. (B9) and (B7).**

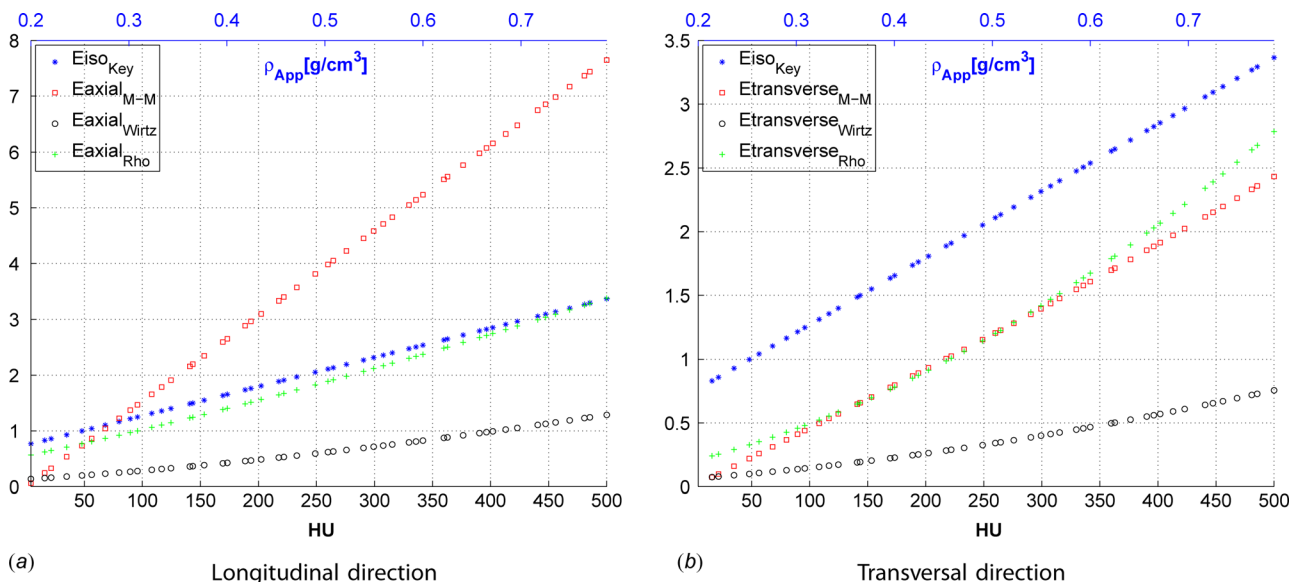


Fig. 9 Young's modulus in different directions versus HU or apparent density for trabecular bone

489 These connections are denoted by the subscript "Rho" in the fol-  
 490 lowing. In Rho the density of cortical bone was determined by  
 491 Archimedes' law and for the trabecular bone by wet weight di-  
 492 vided by volume of the specimens. Poisson ratios are assumed to  
 493 be constant and the shear moduli are calculated as a function of  
 494 Poisson ratio and Young's modulus.

495 **Comparison of the Material Properties Determined**  
 496 **by the Different Methods.** Because Young's modulus has the  
 497 most significant influence on the bone mechanical response, we  
 498 herein illustrate the variation between the different orthotropic  
 499 relations. The mathematical relationships of Young's modulus  
 500 in different directions (longitudinal and transversal) as a func-  
 501 tion of HU or apparent density and the values obtained using  
 502 MM methods are presented in Figs. 8 and 9 for cortical and tra-  
 503 becular regions, respectively. The Young's modulus from the  
 504 isotropic model [36] is also presented. The density is taken as  
 505 the apparent density computed as a linear relation to HU (for  
 506 comparison only).

507 To conclude, well-known elasticity-density relationships for  
 508 orthotropic materials were selected and normalized according to  
 509 bone density and HU. The results show that the values extracted  
 510 using micromechanics (MM) methods are in the range of the  
 511 selected relations more close to the elasticity-density relationships  
 512 suggested by Ref. [58]. The elasticity-density relationships sug-  
 513 gested by Wirtz et al. [20] and used in many studies to investigate  
 514 the influence of anisotropic in FE models yield significantly lower  
 515 values in Young's modulus. The variability in the material values  
 516 was expected, as already noticed for isotropic elasticity-density  
 517 relationships.

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